

Design and potential of instrumented ultramicrotomy

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Ultramicrotomes are generally used for preparation of very thin sections for transmission electron microscopy. Recently it has been shown that when the sample holder of the ultramicrotome is instrumented with a force transducer, it is possible to measure the very small sectioning force during sectioning, and calculate the energy dissipated. In the present work, the instrumentation is further improved. The new sample holder, which uses two piezo-electric force transducers can measure two force components simultaneously. It is not only robust and stiff, but it also shows high sensitivity and reproducibility. It is possible to detect sectioning forces lower than 0.1 mN. The method is demonstrated on two amorphous polymers, poly(methyl methacrylate) and epoxy. Fracture energies in the same order of magnitude as theoretical predictions from chemical bond fracture only are recorded. It is therefore suggested that the method of instrumented ultramicrotomy is a useful tool when information on covalent bond density is needed. Potential future applications are identified including research on nano-scale fracture, characterization of molecular anisotropy and developments of the ultramicrotome. (C) 1997 Elsevier Science Ltd.

(Keywords: ultramicrotome; sectioning forces; fracture)

INTRODUCTION

An ultramicrotome is used for sample preparation for transmission electron microscopy (TEM). By use of a very sharp knife (diamond or glass), extremely thin sections or 'chips' of, for example, polymers or biological materials are prepared. The chip thickness is typically between 30 nm and 150 nm, and in order to get good sectioning results the sample needs to be in the solid, or preferably in the glassy state.

Since microtome sectioning creates two new surfaces one may look at the sectioning as a crack propagation process. Ericson and Lindberg¹ showed that if the ultramicrotome is instrumented in order to measure sectioning forces it is possible to calculate the sectioning energy during the crack propagation. Since the thickness of the sections are very small, this might give us some insight in very local fracture energy dissipation mechanisms.

Microtomes, which are used for section thicknesses from 0.5 to $10 \,\mu$ m, have been instrumented by several workers. Vincent *et al.*²⁻⁵ used a rotary microtome fitted with a modified knife holder which was a load cell. They measured cutting forces, and when the thickness of the sections were decreased continuously from 1 mm to 1 μ m the cutting energy also decreased for a number of biological materials. Atkins and Vincent² suggested that the sectioning energy extrapolated to zero thickness is equal to the critical fracture energy, $G_{\rm C}$. No measurements on polymers have been published, but an interesting study on meat was published by Dobraszczyk *et al.*⁶. Wool and Rockhill⁷ studied the molecular degradation during microtome sectioning by viscometry. Saubermann *et al.*⁸ presented an alternative method to evaluate the sectioning forces on a microtome. However, as seen in the work of Ericson and Lindberg¹ for ultramicrotomes, it is impossible to extrapolate sectioning energies from experiments at the micrometre scale down to zero thickness without significant error. The reason is the dramatic nonlinearity in sectioning energy vs chip thickness in the thickness region below 200 nm. Therefore, in order to measure energy dissipation at a true microscopic scale, one needs to instrument an ultramicrotome.

Several attempts to instrument an ultramicrotome have also been presented in the literature 9-13, all with the purpose to determine sectioning forces. It is found that the techniques for instrumentation vary as much as the reasons for instrumentation. Hodson and Marshall¹⁰ studied the thawing of frozen material during sectioning. They calculated the forces acting on the substrate by measuring the velocity of the specimen arm by a method of electromagnetic induction. When they knew the mass and the retardation of the arm it was possible to calculate the energy dissipated into the material during sectioning. Wikefeldt¹¹ placed two piezoelectric elements in the drive arm of the ultramicrotome. It was then possible to measure both forces acting perpendicular and parallel to the sectioning direction. The reason for his instrumentation was to record instability vibrations, so called 'chatter',

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developed during sectioning. In his thesis he also sectioned poly(methyl methacrylate) (PMMA) and aluminium, and found very small sectioning forces for small chip thicknesses. Helander¹² measured forces during ultramicrotome sectioning at varying sectioning conditions for epoxy. He instrumented the knife holder with two piezoelectric force transducers and reported forces corresponding to sectioning energies around $20-50 \,\mathrm{J\,m^{+2}}$, depending on the chip thickness.

Ericson and Lindberg⁹ instrumented the sample holder of an ultramicrotome. We showed that an ultramicrotome could be used for measurement of very small sectioning forces in order to calculate the energy dissipation during crack growth in amorphous polymers. The energy dissipated for small chip thicknesses was almost in the same order of magnitude as theoretical predictions based on chemical bond fracture only. This suggests that the method is applicable for nano-scale fracture studies. We used a single piezoelectric force transducer mounted on the sample holder and were able to measure forces in the sectioning direction. However, to increase the sample-to-sample reproducibility and learn more about the chip forming process, we need a new stiffer sample holder, able to measure forces in two directions simultaneously.

The objective of this work is to further develop the instrumented ultramicrotome in order to obtain improved sample-to-sample reproducibility, and higher stiffness of the sample holder, and to measure two force components simultaneously. A new sample holder is designed, tested and evaluated. The method of instrumented ultramicrotomy is demonstrated by tests on two amorphous polymers. PMMA and an epoxy. diglycidyl ether of bisphenol A with curing agent diethylamine trianine (DGEBA/DETA). Potential applications of instrumented ultramicrotomy are suggested.

EXPERIMENTAL

Instrumentation of the sample holder

It was earlier suggested¹ that two force components are needed in order to study the mechanics of ultramicrotomy. A special instrumented sample holder was therefore designed and built. Two piezoelectric load cells (Model PCB[®] M209A12 from PCB[®] Piezoelectronics, USA) with battery-powered signal conditioners (Model 480E09 from PCB[®]) were mounted between a sample clamping unit and the connection for the ultramicrotome specimen arm. (See the schematic figure and photo in *Figure 1.*) The analogue signals from the signal conditioner were sampled by a computer for further analysis. The two load cells can only measure compressive forces, and were therefore precompressed to approximately 100 N (each) by two pre-stress rods made of a unidirectional glass fibre/epoxy composite.

Calibration of the sample holder was performed by simulating loads as weights in thin strings, fastened where the sample normally is clamped. Loads from 6 mN to 215 mN were used in different loading angles (simulating different sectioning angles). The two signals from the loads cells were then 'calibrated' within the computer so that the two theoretical force components were continuously calculated and sampled. After calibration, the two pre-stress nuts were secured by an epoxy adhesive. The two analogous signals from the signal

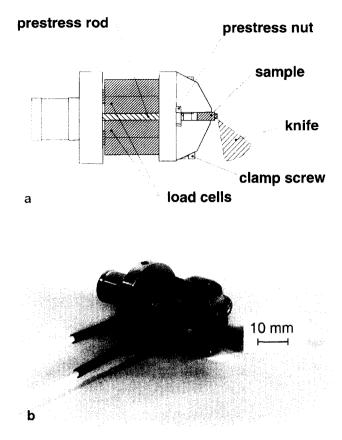


Figure 1 The instrumented sample holder for ultramicrotome sectioning. (a) Schematic figure. (b) Photo

conditioners were then continuously processed in the computer to give the calibrated response.

Materials

For demonstration of the method, two polymers, PMMA and DGEBA/DETA, were sectioned with the new instrumented sample holder. PMMA is a thermoplastic polymer whereas DGEBA/DETA is a thermoset (crosslinked). Both materials are amorphous and in the glassy state at room temperature. For more details about the materials, see ref. 1.

Sectioning procedure

The ultramicrotome used was a LKB 2088 Ultrotome^R V. The knife was a diamond knife (from Juniper Ultra Micro, Sweden). The radius of the knife edge is, according to the manufacturer, between 5 nm and 7 nm. Specimens were first cut from sheets and then trimmed in the ultramicrotome in order to get an appropriate sectioning block, a so called 'mesa', of about 1 mm². The chip thickness, h, was determined from the interterence colour created in the sections on the water surface in the water trough according to *Table 1*. The sectioning speed during measurements was 1 mm s⁻¹ and h was varied between 60 and 250 nm at four discrete levels. For each h, sectioning forces for at least ten sections were recorded. The total work to section per created surface area, W_s , was determined as:

$$W_{\rm s} = \frac{F_{\rm s}}{b} \tag{1}$$

where F_s is the sectioning force component in the sectioning direction (see *Figure 2*), and *b* is the chip width.

Interference colour	Chip thickness, h (nm)	
	$\begin{array}{l} \mathbf{PMMA}\\ (n=1.5)^a \end{array}$	Epoxy $(n = 1.6)^a$
Silvergrey	50-70	50-75
Silver	70-100	75-107
Gold	100-130	107-140
Violet	130-180	140-190
Blue	180-240	190-255

 Table 1
 Interpretation of interface colours of polymer chips on water surface calculated after Patzelt¹⁴

^a n is the refractive index

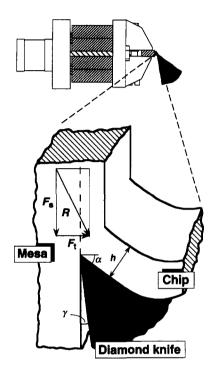


Figure 2 Schematic figure of the ultramicrotome sectioning and the two force components acting upon the sample. F_s is the sectioning force, F_t the transverse force and R the resulting force

RESULTS AND DISCUSSION

Performance of instrumented sample holder

The new instrumented sample holder was not only stiff and very sensitive, but also linear in response to loads in a wider range than needed for ultramicrotome sectioning.

Although the load cells were pre-stressed, the large difference in stiffness between the load cells and the prestress rods (approximately $350 \text{ N} \,\mu\text{m}^{-1}$ and $5 \text{ N} \,\mu\text{m}^{-1}$ respectively) made it possible to fully use the very high sensitivity of the load cells. Since the load cells are positioned symmetrically in the sample holder, it is possible to simultaneously detect both the sectional force, F_s , and the transverse force F_t (see Figure 2). F_t is proportional to the sum of the forces in the two load cells whereas F_s is proportional to the difference. The smallest possible load to detect for each load cell is 0.22 mN. Due to the built-in lever mechanism, the smallest possible detectable F_s and F_t are as low as 0.05 mN and 0.4 mNrespectively. Since the smallest sectioning force per section width for amorphous polymers was found to be about $10 \text{ mN mm}^{-1.1,9}$, the sensitivity of the new sample holder is more than sufficient for the desired purposes. Since the maximum allowable load is ± 200 N, the sample holder is believed to be very robust. After calibration the

sample holder showed very good agreement with simulated loads. Results from the calibration are shown in *Figure 3*, where two different resulting forces, R (13.5 mN and 215 mN) were applied on the specimen holder. When the loading angle, ϕ , is changed, the system give both F_s and F_t simultaneously. The solid lines are the theoretical values, and the data points are experimental values. The correspondence between simulated loads and recorded values is very good for all loads, R(up to 200 mN) and loading angles, ϕ , tested.

To conclude, the new sample holder shows several advantages compared with the sample holder described in refs 1 and 9. Firstly, the earlier sample holder had to be taken apart between a change of samples. The pre-stress in the load cell was then altered and therefore also its response, resulting in poor sample-to-sample reproducibility. The new sample holder is never dismounted and therefore does not suffer from this. Secondly, by use of two load cells, it is possible to detect two force components simultaneously, with much higher sensitivity on a wide load range (from 0.2 mN to much higher than the tested 200 mN). It should therefore also be suited for microtome measurements. Thirdly, the new sample holder is much stiffer. The load cells used here are almost 100 times stiffer than the old sample holder

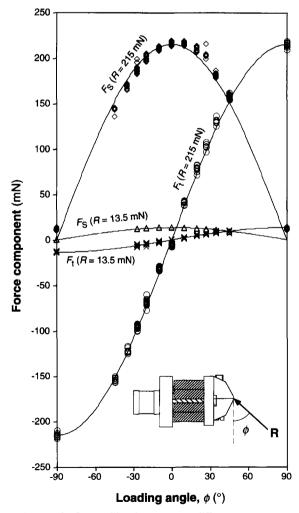


Figure 3 Results from calibration with two different resulting forces, R (13.5mN and 215mN) acting in different loading angles, ϕ , on the specimen holder. For each loading angle two force components, F_s and F_t , are given from the instrumented sample holder. Solid lines are expected values

described in refs 1 and 9, 350 MN m^{-1} compared with 4 MN m^{-1} . This will make it less inclined to chatter, and easier to handle.

Sectioning forces and work to section

In Figure 4 typical sectioning force signals for DGEBA/DETA are presented. Both F_s and F_t are more or less constant over the load pulse. The decrease in both F_s and F_t with time is due to the decay of the signal from piezoelectric load cells. This is a characteristic feature of the load cells, and since it is the variation in load when entering or leaving the sample that is of interest, it is not a problem here. The fluctuations in the forces during sectioning, as seen in Figure 4, could be chatter as well as material variations, or a result from discontinuous crack propagation.

The sectioning force per unit width, F_s/b , which is equal to the work to section (W_s) , and transverse force.

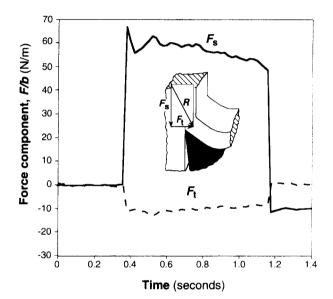


Figure 4 Typical output of F_s and F_t during sectioning of DGEBA DETA with $h \approx 220$ nm. Sampling frequency is 50 Hz and nominal sectioning speed is about 1 mm s⁻¹

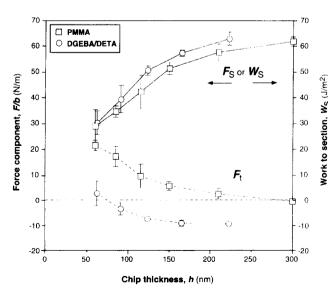


Figure 5 Work to section, W_s , and sectioning force components, F_s/b and F_t/b for PMMA and DGEBA/DETA. The sectioning speed is 1 mm s⁻¹

 F_t/b , as a function of section thickness h is presented in Figure 5 for PMMA and DGEBA/DETA. As observed previously¹, W_s decreases down to less than $30 \,\mathrm{Jm^{-2}}$ for the two materials. Keeping in mind that the macroscopically determined fracture energy, G_{ic} , for PMMA and DGEBA/DETA is $500 \text{ Jm}^{-2.15}$ and $130 \text{ Jm}^{-2.16}$, respectively, it is remarkable that the W_s values for the two polymers do not differ more. The theoretically determined minimum fracture energy, the so-called intrinsic fracture energy, G_0 , is about 1 J m⁻² for PMMA¹⁷. G_0 is the energy required to cut the chemical bonds at the fracture plane, and is believed not to differ significantly for DGEBA/DETA. Since W_s for very thin sections and G_0 are in the same order of magnitude, we believe that the total energy dissipated during sectioning is dominated by chain scission. Other possible energy dissipation mechanisms include chain stretching prior to scission and chain pull-out after bond rupture¹⁸ However F_t/b differs significantly between PMMA and DGEBA/DETA. We believe that differences in other energy dissipated mechanisms, such as friction, may cause this. In order to estimate this effect of other energy dissipation mechanisms, one needs to model the chip formation process and the frictional forces. This is done elsewhere by Ericson and Lindberg²⁰ for five different amorphous polymers.

Potential applications for instrumented ultramicrotomy

Results from instrumented ultramicrotome studies are encouraging. We strongly believe that instrumented ultramicrotomy opens room for several new applications for ultramicrotomes, for material properties characterization as well as for sectioning optimization (sectioning parameters or the equipment). A number of potential applications are presented below.

- 1. Nano-scale crack growth studies. Since the measured work to section, W_s , is in the same order of magnitude as theoretical predictions based on chemical bond fracture only, we believe that the method is applicable to research on nano-scale crack growth. In addition to fundamental research on molecular fracture, this might be useful in the development of new toughened materials. However, the crack propagation processes and the chip forming process are at present not very well understood. An increased insight in the different energy dissipation mechanisms and the deformation mechanisms are needed.
- 2. Characterization of molecular anisotropy. Since the energy dissipated is proportional to the density of covalent bonds, the method is also sensitive to molecular anisotropy for macromolecular materials such as polymers. Ericson and Lindberg²⁰ showed that W_s varied significantly when a highly oriented amorphous polymer was sectioned in different directions. The variation in W_s corresponded well to variations in Young's modulus and dimensional recovery after heat treatment.
- 3. Determination of cell wall fraction in cellular solids. Since W_s is proportional to the sectioned area, it may be possible to quantify the volume fraction of cell walls in cellular materials.
- 4. Optimization of ultramicrotome sectioning parameters. In order to get high quality sections with an ultramicrotome, one has to set several sectioning parameters, such as sectioning speed, rake angle of

the knife etc. We believe that the best quality implies a minimum of sectioning energy. This was also suggested for microtomes by Willis²¹ and Atkins and Vincent² and Willis and Vincent⁵. Therefore it may be possible to set optimum sectioning parameters from measurements of dissipation energies for different setting for sectioning parameters. Mechanical chatter can also be detected as force variations, as shown by Wikefeldt¹¹.

- 5. Knife sharpness. As shown by Ericson and Lindberg²⁰, it is possible to detect when the knife is dull. For a dull knife the W_s does not decrease significantly for small h.
- 6. Measurement of chip thickness. Since the work to section strongly depends on h (especially for small h, see Figure 5) one can read the chip thickness from the level of W_s if the relation between h and W_s is known. This may be useful when the material is not transparent, for example aluminium, or when the sections are so thin that they do not give any interference colour at the water surface.
- 7. Automation of knife approach. One critical moment during ultramicrotome sectioning is the approach of the very sharp and sensitive knife edge onto the sample. The knife is easily damaged if the first contact with the sample is not very gentle. Since the sample holder easily detects forces in the mN range, the approach could be motorized, and stopped when a critical force is exceeded.

CONCLUSIONS

Developments in the design of the instrumented ultramicrotome resulted in significant improvements. By use of two piezoelectric load cells in the sample holder it was possible to measure two force components simultaneously during sectioning. The new instrumented sample holder showed improved sample-to-sample reproducibility and sensitivity. The smallest load that was possible to detect was less than 0.1 mN. The method was successfully demonstrated by sectioning of PMMA and an epoxy (DGEBA/DETA). Very small levels of work to section were measured (less than $30 \,\mathrm{J\,cm^{-2}}$) for both materials. It is suggested that the method of instrumented ultramicrotomy may be useful whenever information on chemical bond density in a material is of interest. For instance, fracture energies at the molecular level and molecular anisotropy can be studied with the method. Furthermore, developments such as in situ chip thickness determination and an automatic knife approach are possible with the new sample holder.

ACKNOWLEDGEMENTS

This work was mainly funded by the faculty of Luleå University of Technology. Equipment was funded by the Swedish Research Council for Engineering Sciences (TFR).

REFERENCES

- 1. Ericson, M. L. and Lindberg, H., Journal of Materials Science, 1996, 31, 655.
- 2. Atkins, A. G. and Vincent, J. F. V., Journal of Materials Science Letters, 1984, 3, 310.
- 3. Vincent, J. F. V., European Microscopy and Analysis, 1991, May, 13.
- 4. Allison, R. T. and Vincent, J. F. V., Journal of Microscopy, 1990, 159, 203.
- 5. Willis, A. and Vincent, J. F. V., *Journal of Microscopy*, 1995, **178**, 56.
- 6. Dobraszczyk, B. J., Atkins, A. G., Jeronimidis, G. and Purslow, P. P., *Meat Science*, 1987, **27**, 25.
- 7. Wool, R. P. and Rockhill, A. T., J. Macromol. Sci.-Phys., 1981, **B20**(1), 85.
- 8. Saubermann, A. J., Riley, W. D. and Beeuwkes III, R., Journal of Microscopy, 1977, 111, 39.
- Ericson, M. L. and Lindberg, H., Polymat '94, 19-22 September 1994, Imperial College, London 1994. Institute of Materials, London, p. 614.
- Hodson, S. and Marshall, H., Journal of Microscopy, 1972, 95, 459.
- 11. Wikefeldt, P., Ph.D. thesis No 86, Chalmers Institute of Technology, Sweden, 1973 (in Swedish).
- 12. Helander, H. F., Journal of Microscopy, 1974, 101, 81.
- 13. Willett, J. L., O'Conner, D. M. and Wool, R. P., J. Polym. Sci., Polym. Phys. Edn. 1986, 24, 2583.
- Patzelt, W. J., Polarisationsmikroskopie. Grundlagen, Instrumente, Anwendungen. Ernst Leitz Wetzlar GMBH, Wetzlar, 1974.
- Mark, H. F., Bikales, N. M., Overberger, C. G., Menges, G. and Kroschwitz, J. I. (ed.), *Encyclopedia of Polymer Science and Engineering*, Vol. 7. John Wiley, New York, 1987, p. 372.
- Asp, L. E., Berglund, L. A. and Gudmundson, P., Composites Science and Technology, 1995, 53, 27.
- 17. Kausch, H. H., *Polymer Fracture*. Springer-Verlag, Berlin, 1987, p. 273.
- 18. Sperling, L. H., *Introduction to Physical Polymer Science*, 2nd edn. John Wiley, New York, 1992, p. 537.
- Sambasivam, M., Klein, A. and Speling, L. H., *Macromolecules*, 1995, 28, 152.
- 20. Ericson, M. L. and Lindberg, H., Mechanics of ultramicrotomy, in preparation.
- Willis, A., Design and development of an instrumented microtome. Ph.D. thesis, Dept of Pure & Applied Zoology, University of Reading, UK, 1988.